This article was downloaded by: On: 19 January 2011 Access details: Access Details: Free Access Publisher Taylor & Francis Informa Ltd Registered in England and Wales Registered Number: 1072954 Registered office: Mortimer House, 37- 41 Mortimer Street, London W1T 3JH, UK



## International Journal of Polymeric Materials

Publication details, including instructions for authors and subscription information: <http://www.informaworld.com/smpp/title~content=t713647664>

## A Correlation Between Micro and Macro Structure of High Cis-Polybutadiene and its Performance in Tyre Compound

Nalini R. Kumar<sup>a</sup>; Arup K. Chandra<sup>a</sup>; R. Mukhopadhyay<sup>a</sup> a Hari Shankar Singhania Elastomer & Tyre Research Institute Jaykaygram, Kankroll, Rajasthan, India

To cite this Article Kumar, Nalini R. , Chandra, Arup K. and Mukhopadhyay, R.(1996) 'A Correlation Between Micro and Macro Structure of High Cis-Polybutadiene and its Performance in Tyre Compound', International Journal of Polymeric Materials, 34: 2, 91  $-$  103

To link to this Article: DOI: 10.1080/00914039608031468 URL: <http://dx.doi.org/10.1080/00914039608031468>

# PLEASE SCROLL DOWN FOR ARTICLE

Full terms and conditions of use:<http://www.informaworld.com/terms-and-conditions-of-access.pdf>

This article may be used for research, teaching and private study purposes. Any substantial or systematic reproduction, re-distribution, re-selling, loan or sub-licensing, systematic supply or distribution in any form to anyone is expressly forbidden.

The publisher does not give any warranty express or implied or make any representation that the contents will be complete or accurate or up to date. The accuracy of any instructions, formulae and drug doses should be independently verified with primary sources. The publisher shall not be liable for any loss, actions, claims, proceedings, demand or costs or damages whatsoever or howsoever caused arising directly or indirectly in connection with or arising out of the use of this material.

# **A** Correlation Between Micro and Macro Structure of High Cis-Polybutadiene and its Performance in Tyre Compound

NALINI R. KUMAR, ARUP K. CHANDRA and R. MUKHOPADHYAY

*Hari Shankar Singhania Elastomer* & *Tyre Research Institute Jaykaygram, Kankroli, Rajasthan 3 13 342, India* 

#### *(Received 20 February 1996)*

High cis-polybutadiene from twelve different sources synthesised by using Ziegler Natta catalyst system based on four different metals viz. Co, Nd, Ni and Ti was studied. Raw polymers were characterised by measuring cis and trans content, vinyl content, branching index, molecular weight, molecular weight distribution, glass transition and melting temperatures, Mooney and delta Mooney values and Mooney relaxation behaviour. Performance of the polymer in compound was evaluated in truck and light truck tyre tread and sidewall compounds as well as in ASTM formulation. Extrusion characteristics of the truck tread compound were determined by Brabender Plasticorder. Stress-strain properties (unaged  $\&$  aged), dynamic mechanical properties, abrasion loss, resilience and heat build up characteristics were measured. **A** correlation between micro and macro structure and optimum performance of polybutadiene was established.

*Keywords:* Polybutadiene; cis content; tread compound; branching index and Mooney relaxation.

## **INTRODUCTION**

Automobile industries are coming up with vehicles having increased speed, fuel efficiency, braking efficiency, comfort, etc. Accordingly the tyres to cope up with these requirements are demanded. This demand is fulfilled by enhanced tyre design and using new improved compounds. As a consequence compounders are looking for more and more selective and specialty rubbers whch will provide desired properties for different components of tyres. One of the mostly used rubbers in automobile tyre is high cis-polybutadiene (BR). This rubber is used in blends with natural rubber and styrene butadiene rubber in many components, such as sidewall, tread, rim chafer strips, etc. The advantages of using polybutadiene are improved wear and cut resistance, resilience, hysteresis, ozone resistance, low temperature flexibility and low heat development under dynamic stress.

At present commercial polybutadienes are usually produced by solution process using Ziegler Natta type catalyst system based on metal organic complexes (e.g. Co, Nd, Ni, Ti, etc.). The micro & macro structure namely cis & trans content, molecular weight, molecular weight distribution, branching of polymer chain, etc. depend on the catalyst system used. These are again responsible for overall performance of the rubber. So it is absolutely necessary to select right type of polybutadiene to have compound with improved performance.

The importance of cis content in solution **BR** was reported by Vohwinkel [ 11. Comparative studies on **BR** of different catalytic origins were reported by various researchers  $\lceil 2 - 4 \rceil$ . Colombo, *et al.* studied tyre performance using a new high cis-polybutadiene *[S].* However, the information which correlates micro and macro structure of **BR** with performance of tyre compounds in current literature is inadequate. This investigation is an endeavour to establish a correlation between catalyst type (which is responsible for producing **BR**  having different micro and macro structure) and overall performance of the **BR** in truck and light truck tyre compounds as well as in ASTM formulation.

## **EXPERIMENTAL**

#### **Materials**

Polybutadiene used were commercial ones, synthesised by using Ziegler Natta catalyst system based on cobalt, neodymium, nickel and titanium. All other ingredients used in this study were from standard sources. In this study  $BR_1$ —BR<sub>6</sub> were based on Co catalyst,  $BR_7$ —BR<sub>9</sub> on Nd,  $BR_{10}$ —BR<sub>11</sub> Ni and **BR,** were based on Ti catalyst systems.

## **Procedure**

## *Characterisation of raw rubber*

Macro structure (cis and vinyl content) was determined by a Perkin Elmer model 1760 **FTR** spectrometer and calculated according to Morero method *[6].* 

The branching index, G was evaluated according to Drott-Mendelson method [7]. This is defined as the ratio between the experimental intrinsic viscosity and the theoretical one obtained from **GPC** chromatograms assuming all molecules to be linear.

Molecular weight and molecular weight distribution were measured by a Multipore-Waters model **712** Wisp (Lichrogel columns array : PS **400000,** PS **40000, PS 400,** PS **40)** and using the following Mark-Houwink constants:

 $K = 0.000457$ :  $\alpha = 0.693$ .

Melting temperature,  $T_m$  and glass transition temperature,  $T_a$  were measured by differential scanning calorimetry method using a DSC 30 model Mettler calorimeter.

Mooney value was determined as per ASTM **D1646** and delta Mooney was measured at 100°C by a Monsanto Mooney viscometer MV2000E.

#### **Mooney relaxation**

The Mooney relaxation for the raw rubber was measured subsequent to the measurement of Mooney value by stopping the rotor immediately after the test time is complete. Depending on the elasticity of the polymer, the torque decreases more or less rapidly. The ratio of the torque after **30 s** stop to the Mooney value at a time of  $1 + 4$  min (in %) is the Mooney relaxation,  $M_R$  $[8, 9].$ 

## **Mixing**

Mixing of compounding ingredients with rubber was carried out in a laboratory banbury of chamber volume 1.5 litre.

## **Extrusion behaviour**

Extrusion behaviour of the truck tyre tread compound was studied by using a Brabender Plasticorder (Model PL 2000 - **3).** The die used was a round capillary having diameter of *5* mm, screw speed was kept constant at **30** RPM and temperature of extrusion was **90°C.** Diameter of the extrudate was measured at the exit of the capillary by an opto-electronic rod thickness tester (supplied by Brabender **OHG,** Germany). Extrudate swell presented here is the ratio of diameter of the extrudate,  $d_e$  to the diameter of the capillary die,  $d_c$ .

## **Testing**

Stress-strain properties of the vulcanizates were measured using a Zwick Universal Testing machine (model 1445) in accordance with ASTM D 412 and D 624 at a cross head speed of 500 mm/min.

Abrasion loss was determined as per DIN 53416.

Loss tangent, tan  $\delta$  of the vulcanizates was measured by using a Dynamic Viscometer, Rheovibron model DDV-111-C.

Resilience was determined by a Resilo-tester according to ASTM D 2632.

Fatigue to failure properties were determined by a Monsanto Fatigue to Failure Tester at an extension of 100%. Heat build up was measured using a Martin's ball fatigue tester at ambient temperature under 15 Kg load after 1950 revolutions.

## **RESULTS AND DISCUSSION**

### **Micro and macro structure**

Micro structure of twelve polybutadiene rubbers is presented in Table I. The table show that the 1,4-cis content is comparable for Co and Nd based BR, but it is slightly lower for Ni based BR and lowest value is observed for Ti-BR. The 1, 2-vinyl content is maximum for Ti-BR followed by Ni, Co and Nd-BR (Tab. I). The lowest vinyl content of BR indicates that Nd-BR have got the highest stereoregularity in the polymer chains. The branching index, G which is a measure of polymer chain linearity i.e. G values close to one means linear polymer whereas the lower the G values higher is the macromolecular

**TABLE I** 



branching. It is clear from the table that chain linearity is maximum for Nd catalyst based BR followed by Ti, Co and Ni.

## **MOLECULAR WEIGHT AND MOLECULAR WEIGHT DISTRIBUTION**

Polybutadiene synthesised by using Ni catalyst system is having broadest molecular weight distribution **(MWD)** followed by the BR based on **Nd,** Co and Ti catalyst systems. The highest molecular weight **(M,)** is observed for Nd based BR (except  $BR_6$  which is a Co-BR) followed by Ni, Co and Ti-BR (Tab. II).  $M_w$  and MWD play important roles to the mechanical properties and processing behaviour of an elastomer [10, 11]. An elastomer with broad MWD generally **shows** good processing behaviour but Nd based BR inspite of their broad **MWD** offer processing difficulties like higher black incorporation time. Higher **M,** and branching index of Nd based BR probably outweigh the advantages of broad **MWD.** 

## **GLASS TRANSITION TEMPERATURE, T, AND MELTING TEMPERATURE, T,**

 $T<sub>o</sub>$  and  $T<sub>m</sub>$  values are comparable for all the BR used in this study (shown in Tab. II). It may be noted that  $T_q$  is strongly related to wear and traction properties of an elastomer [12].



**\*-Does** not **crystallize** 

## **MOONEY AND DELTA MOONEY VALUES**

Mooney value of all the BR studied is comparable but delta Mooney is slightly lower for Nd-BR than that of the others (Tab. III). Lower delta Mooney value indicates lower branching of the polymer chains. This observation is corroborated with the higher values of branching index of Nd<sup>-BR</sup> (Tab. I).

## **MOONEY RELAXATION**

Mooney relaxation is a measure of the elasticity of a polymer. Smaller the value of  $M_{\rm R}$ , lower is the elastic characteristic and better is its processing behaviour  $[10 - 12]$ . The stress relaxation testing can be used for processability quality control test for synthetic rubber since this test is relatively quick than the other rheological tests. The values of  $M_R$  are tabulated in Table III. There is a considerable variation in the value of  $M_R$  for Co-BR though there is no such variation observed in the processability of these BR. For other types of BR,  $M_R$  values were very close for a particular catalyst type. It is to be noted that there is a correlation between the values of delta Mooney and  $M_R$  i.e. if delta Mooney is high for a particular BR  $M_R$  is also high for that BR e.g. BR<sub>2</sub>. **A** similar observation was also reported by other researcher **[13].** 

#### **EXTRUSION BEHAVIOUR**

Extrudate swell of the truck tread compound does not show noticeable difference for the BR having different catalytic origins (Tab. **IV).** Extrusion rate for this compound also maintained similar behaviour. The extrudate surface was found smooth for all the BR except  $Nd$ —BR. This may be because of higher  $M_w$  of this type of BR. These observations clearly indicate that extrudate swell and extrusion rate do not depend on the catalyst system but nature of the extrudate surface does get affected by the same.

## **VULCANIZATE PROPERTIES**

**As** shown in Figure 1 resilience property of the vulcanizates in ASTM formulation is comparable for all the BR used here.

It has been established by several scientists that loss tangent of tread compounds at  $50 - 100$ <sup>o</sup>C is directly proportional to the rolling resistance





**M, is Mooney Relaxation** 

**TABLE IV Formulations** 

<i>Ingredients</i>		Truck tread Lt. Truck tread	Sidewall	
NR	40		70	
BR	60	30	30	
<b>SBR</b>		70	-	
<b>Black</b>	60	65	45	
Curing System	C۷	C٧	CV	

**NR-Natural rubber** 

**BR-Polybutadiene rubber** 

**SBR-Styrene butadiene rubber** 

**CV-Conventional vulcanization** 

[14,15]. The loss tangent values (tan  $\delta$ ) at 11 Hz and at three different temperatures viz., 25", 50" and **75°C** of the vulcanizates of ASTM and tread compound are presented in Figures **2** & 3. In the case of tread compound Ni **(BR,,** & **BR,,)** and Ti **(BR,,)** based **BR** have highest loss tangent value followed by Co and Nd based **BR** at all the temperatures studied. The trend remained similar also for the ASTM vulcanizates. It is clear from the figures that there is a systematic drop in the loss tangent value as the temperature increases. This may be due to the increase in the mobility of polymer chain with increase in temperature which reduces the resistance to molecular motion  $[16]$ 

Stress-strain properties of the vulcanizates of truck tyre tread compound are tabulated in TableV. It is clear from the table that modulus at 300% elongation is higher for **Co-BR** than the other type of **BR.** The trend



FIGURE 1 **Rebound** resilience values of the vulcanizates of ASTM formulation.



FIGURE 2 Tangent delta values of the vulcanizates of ASTM formulation.

remained similar after aging for **4** weeks at 70 'C. In the case of tensile strength  $(T, S)$  unaged values are comparable for all the BR but  $Co$ —BR show slightly higher T. **S.** than that of the other types of BR after aging for 3 days at 105°C and 70<sup>o</sup>C. Elongation at break and tear resistance properties are comparable before and after aging for all the BR used here. Retention of physical properties are important for tyre compounds from the service life point of view. Higher the retention better is the service life.



FIGURE 3 Tangent delta values of the vulcanizates **of** truck tread compound,

Abrasion loss which is one of the most important properties for tread compound when investigated in light truck tread formulation all the BR showed a comparable value except  $BR_{10}$  (Ni-BR) which showed highest loss but BR<sub>12</sub> (Ti--BR) the lowest value (Fig. 4). The same for the truck tread showed a comparable abrasion loss for all the BR studied here. The abrasion

TABLE **V** 



Downloaded At: 11:51 19 January 2011 Downloaded At: 11:51 19 January 2011

BR type BR,	Modulus at 300 $\%$ Elongation (MPa)		Tensile Strength (MPa)		<b>Elongation</b> at Break $\binom{o}{o}$		Tear Resistance (N/mm)				
	9.5		13.2	19.6	9.5	17.6	525	225	400	78	20 41
			(139)		(48)	(90)		(43)	(76)		$(26)$ $(53)$
BR <sub>2</sub>	8.6		12.5	18.9	9.9	16.5	550	225	400	64	52 19
			(145)		(52)	(87)		(41)	(73)		(30) (81)
BR,	9.6	$\overline{\phantom{0}}$	13.4	19.5	10.3	17.6	525	225	400	78	20 40
			(140)		(53)	(90)		(43)	(76)		$(26)$ $(51)$
BR <sub>4</sub>	9.5	$\overline{\phantom{a}}$	13.7	19.6	9.8	16.6	525	200	350	65	20 42
			(144)		(50)	(85)		(38)	(67)		$(31)$ $(65)$
BR <sub>5</sub>	9.1	-	12.9	20.7	9.6	17.5	575	225	400	51	20 40
			(142)		(46)	(85)		(39)	(70)		(39) (78)
BR <sub>6</sub>	9.3	$\rightarrow$	13.2	19.8	10.0	17.3	510	220	390	65	22 41
			(142)		(51)	(87)		(43)	(76)		$(34)$ $(65)$
BR <sub>7</sub>	8.1		11.7	18.8	9.6	15.7	575	250	375	68	21 41
			(114)		(51)	(84)		(43)	(65)		$(31)$ $(60)$
BR <sub>s</sub>	8.4	$\rightarrow$	11.7	19.0	9.6	16.5	550	250	400	52	19 40
			(139)		(51)	(87)		(45)	(73)		(37) (77)
BR <sub>o</sub>	8.2		11.8	18.7	9.5	15.9	580	245	380	70	22 43
			(144)		(51)	(85)		(42)	(66)		$(31)$ $(61)$
$\mathbf{BR}_{10}$	7.6		11.1	18.7	9.0	16.1	600	225	400	61	18 53
			(146)		(48)	(86)		(38)	(67)		(30) (87)
$\mathbf{BR}_{11}^-$	8.7		11.9	19.5	10.2	15.4	575	250	350	66	29 47
			(137)		(52)	(79)		(43)	(61)		(44) (71)
$\mathbf{BR}_{12}$	6.9		11.6	19.1	9.9	16.3	625	225	400	76	20 38
			(168)		(51)	(85)		(36)	(64)		$(26)$ $(50)$

**TABLE VI Stress-strain Properties of Truck Tread Compound** 

For **each property the second and third columns indicate the values aged for** 3 **days at** 105 **"C and** for 4 **weeks at** 70 *-C,* **respectively. The figure within the parenthesis indicates per cent retention** of **the property** 



**FIGURE** 4 **Abrasion** loss **values of Truck and light Truck tread compounds.** 

process can be described as detachment of rubber particles from the bulk due to catastrophic tearing of the rubber surface [17].

Fatigue to failure behaviour of the vulcanizates of truck tread, light truck tread and sidewall compounds is presented in Figure *5.* The figure shows that truck tread with Co-BR fatigue property is better than that of the other except  $BR_3$  (a Co-BR) which required minimum cycles to fail. In the case of light truck tread,  $Ti-BR (BR<sub>2</sub>)$  required maximum cycles to fail followed by Co, Ni and Nd - BR. But in sidewall compound Nd-BR showed maximum  $(BR_7)$  and minimum  $(BR<sub>8</sub>)$  fatigue resistance property but for other types of BR it is comparable. There are reports available in literature which state that BR based on Nd catalyst systems has got improved fatigue properties over the other catalyst based BR *[S,* 181. This observation was not reflected in the present study.

Heat build up values of truck tread compound are shown in Figure 6. This shows that the temperature development varies from  $57^{\circ} - 66^{\circ}$ C, where lower





**FIGURE 6 Heat build up values of the tread compound in Martin's ball fatigue test.** 

values are observed for Nd and  $Co$ —BR but the higher are for Ni and  $Ti-BR$ . Heat development and effect of dynamic stress on the vulcanizate play an important role towards the performance and life of a tyre. Particularly Martin's ball technique in which heat development takes place because of shearing and compression of the sample which exactly simulate the service conditions of tread compounds.

## **CONCLUSIONS**

The 1,4 cis content of polybutadienes synthesised by using  $\cos \theta$  Nd catalyst systems is found to be higher than that of  $Ni-BR$  and  $Ti-BR$ .

The order of stereoregularity is as follows:

 $Nd-BR$  $Ti-BR$  $Ni-BR$  $Co-BR$ .

The broadest MWD is found for  $Ni-BR$  followed by Nd, Co and Ti-BR.

Mooney value is comparable for all the BR used in this study. Delta Mooney and Mooney relaxation values are higher for few Co-BR and  $Ti-BR$  than the other type of BR.

Extrudate swell and extrusion rate are almost same for all the BR used here. Nd-BR showed rough extrudate surface whereas other types of BR showed smooth surface.

The highest loss tangent value is observed for  $Ni$ -BR and  $Ti$ -BR followed by Co and Nd based BR.

Unaged stress-strain properties are better for Co based BR than that of the others. The same after hot air aging are comparable.

Abrasion loss is found to be comparable for all the BR studied.  $Co$ —BR showed better fatigue properties and a comparable heat development.

The polybutadiene based on Co catalyst system showed balanced processing and physical properties in tyre compounds as well as in ASTM formulation.

### *Acknowledgments*

The authors are thankful to **Mr.** F. Mistrali, Enichem Elastomers, Italy for extending help in determining the micro & macro structure of polybutadiene rubber.

#### *References*

- [l] Vohwinkel, K. (1993) Kautsch Gummi. Kunstst, *46* (6), 458.
- [2] Chakravarty, S. N., Mustafi, **S.** K. and Mukherjee, A. K. (1989) Rubber News, 28 (9), 33.
- [3] Chakravarty, S. N. and Mustafi, S. K. (1991) J.IRI **1 (2),** 13.
- [4] Lauretti, E., Miani, B., Mistrali, F. (1993) Tire Tech. International, 72.
- [5] Colombo, L. Buetti, *S.* Di Pasquale A. and Miami, *8.* (1993) Kautsch Gummi. Kunstst, 46 (6), 458.
- [6] Morero, D., Santambrogio, A., Porn, L., Ciampelli, F. (1959) Chim. Ind. (Milan) 41, 758.
- [7] Drott, E. E. Mendelson, R. A. (1970) *J. Polyrn Sci.,* **A2,8,** 1361.
- [S] Koopmann, R. and Kramer, H. (1982) ASTM-Symposium on processability of rubber, Toronto, Canada, June 21,.
- [9] Kramer, H. and Koopmann, R. (1984) Kautsch Gummi. Kunstst, 37, 10
- [10] Breemhaar, W., Koopman, R., Markert, J. and Noordermeer, J. (1993) Kautsch Gummi. Kunstst, 46 (12), 957.
- [11] Male, F. J. (1994) Rubber world, August, 73.
- [12] Gargani, L. and Bruzzone M. Fatigue Resistance of Polybutadiene and Effect of Micro structure: Enichem Elastomen **S** P A, Italy.
- [13] Stollfuss, D. I. Use of Polybutadiene in Tyre Manufacture with special reference to BUNACB grades: Presented at **14th** IRMRA Rubber Conference, Madras, India.
- [14] Hong, **S.** W. and Birdsall, D. A. (1990) Presented at the ACS meeting at Las Vegas, Nevada, May 29.
- [lS] Hong, **S.** W. (1994) Presented at the National Rubber Conference, November 14-15, Cochin, India.
- [ 161 Laube, S. G. 'Dynamic Properties a Fundamental Approach'-A Technical Report, TG-77-2, Cabot Corporation (Carbon Black Division), Massachusetts, USA.
- [17] Grosh, K. A. Shallamach, A. (1966) Rubber Chem. Techno]., 39.287.
- [IS] Lauretti, **E.,** Miani, B. and Mistrali, F. (May 1994) Rubber World, 34.